

# Activation of additional energy dissipation processes in the magnetization dynamics of epitaxial chromium dioxide films

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(Received 2 October 2007; revised manuscript received 14 December 2007; published 29 January 2008)

The precessional magnetization dynamics of a chromium dioxide (100) film is examined in an all-optical pump-probe setup. The frequency dependence on the external field is used to extract the uniaxial in-plane anisotropy constant. The damping shows a strong dependence on the frequency, but also on the laser pump fluence, which is revealed as an important experimental parameter in this work: above a certain threshold further channels of energy dissipation open and the damping increases sharply. This behavior might stem from spin-wave instabilities.

DOI: [10.1103/PhysRevB.77.020412](https://doi.org/10.1103/PhysRevB.77.020412)

PACS number(s): 75.30.Gw, 76.90.+d

As a predicted half-metallic ferromagnet,<sup>1</sup> chromium dioxide (CrO<sub>2</sub>) shows a spin polarization at the Fermi level that comes close to full polarization,<sup>2–4</sup> which is the defining property of a half metal. Therefore, CrO<sub>2</sub> has attracted a lot of interest as a possible material for future spintronic devices.<sup>5,6</sup> To achieve high processing speed in such devices, a fundamental insight into the magnetization dynamics of CrO<sub>2</sub> is needed.

An all-optical pump-probe setup utilizing femtosecond laser pulses<sup>7,8</sup> [time-resolved magneto-optical Kerr effect (TRMOKE)] allows investigation of the magnetization dynamics of ferromagnetic films in the time domain. In this setup, the laser-induced demagnetization<sup>9,10</sup> and subsequent remagnetization<sup>11</sup> are accompanied by a change of the equilibrium direction of magnetization, which can be understood as a picosecond field pulse<sup>12</sup> that leads to precessional motion according to the Landau-Lifschitz-Gilbert (LLG) equation<sup>13,14</sup>

$$\frac{d}{dt}\mathbf{M} = -\gamma_0\mathbf{M} \times \mathbf{H}_{\text{eff}} - \frac{\alpha}{M}\mathbf{M} \times \frac{d}{dt}\mathbf{M}, \quad (1)$$

where  $\gamma_0 = \mu_0|\gamma|$  and the dimensionless parameter  $\alpha$  accounts for the damping of the magnetic motion. This damping term is derived by introducing isotropic Rayleigh-like energy dissipation. In general, the microscopic mechanism for damping does not obey these assumptions. Nevertheless, the precessional motion can still be described by an effective and possibly frequency-dependent damping parameter  $\alpha_{\text{eff}}$ . It has been shown that this parameter can be extracted from the precessional motion traced in an all-optical TRMOKE setup.<sup>8,15</sup> In these experiments, the equilibrium magnetization is canted out of the film plane so that the laser-induced demagnetization comes along with a change of the direction of the shape anisotropy field of the sample. Zhang *et al.* have demonstrated<sup>16</sup> that the in-plane anisotropy of a CrO<sub>2</sub>(100) film can be utilized in an optical pump-probe setup for the generation of an in-plane anisotropy field pulse that induces precessional motion. Recently, several TRMOKE experiments with a similar configuration were reported.<sup>17–20</sup> Here,

we present a systematic all-optical measurement of the precessional frequency and damping of a 300 nm CrO<sub>2</sub>(100) film. The examined pump fluence dependence of the sample shows the opening of an additional channel of energy dissipation at a sufficiently high perturbation from the equilibrium configuration.

The CrO<sub>2</sub> film is grown epitaxially by chemical vapor deposition on a TiO<sub>2</sub>(100) (rutile) substrate.<sup>21</sup> The 300-nm-thick film, examined in detail for this work, is expected to show uniaxial magnetic in-plane anisotropy with the [001] axis being the in-plane easy axis and an effective first-order anisotropy constant of  $15 \times 10^3 \text{ J/m}^3$  at room temperature.<sup>21</sup> This anisotropy originates from an interplay between crystalline and strain-induced magnetic anisotropy. With the external field applied in the in-plane hard-axis direction, the free energy density of the system can be written as

$$F = -\mu_0 M_s H_{\text{ext}} \sin \phi \sin \theta + \frac{1}{2} \mu_0 M_s^2 \cos^2 \theta + K_{1,\text{eff}} \sin^2 \phi, \quad (2)$$

where the first term represents the Zeeman energy, and contributions from the shape and the in-plane anisotropy are taken into account by the second and the third terms, respectively. The angles  $\phi$  and  $\theta$  are named according to Fig. 1(a).

In our experiment, the probe and pump pulse are generated by a titanium:sapphire femtosecond laser together with a regenerative amplifier (repetition rate 250 kHz, pulse width 60 fs, spot sizes of probe and pump pulses 40 and 60  $\mu\text{m}$ , respectively). We utilize a double modulation scheme so that the polarization of the probe pulse is modulated with a photoelastic modulator and the intensity of the pump pulse with a mechanical chopper. The sample is mounted at room temperature with the external magnetic field in the plane of incidence parallel to the hard axis [Fig. 1(a)]. It has been shown that the hysteresis curve for this sample in this configuration can be explained by coherent rotation of a single domain.<sup>22</sup> Therefore, the equilibrium configuration, i.e., the azimuthal angle  $\phi$ , is given by the minimum of the free

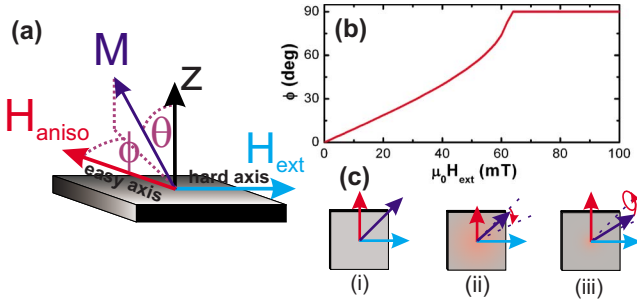


FIG. 1. (Color online) (a) Configuration of the sample system: The external field is applied in the hard-axis direction; the angles  $\theta$  and  $\phi$  are chosen according to Eq. (2). (b)  $\phi$  calculated from Eq. (2) with constants as given in the text. (c) In-plane anisotropy field pulse: the impinging pump pulse changes the equilibrium configuration (i) due to lattice heating and adjustment of the magnetic anisotropy on the time scale of 1 ps (ii); the slow recovery of the initial configuration is accompanied by precessional motion (iii).

energy in Eq. (2). In Fig. 1(b),  $\phi$  is depicted as a function of the external field where the room temperature material parameters are chosen to be  $K_{1,\text{eff}} = 15 \times 10^3 \text{ J/m}^3$  and  $\mu_0 M_s = 603 \text{ mT}$ .<sup>23</sup> The laser-induced demagnetization in  $\text{CrO}_2$  reaches its maximum after 200–300 ps, in contrast to, e.g., nickel, where the demagnetization occurs on a time scale of hundreds of femtoseconds. This behavior has been attributed to the half-metallic character of chromium dioxide and a resulting decoupling between the spin and the electron system.<sup>24</sup> Nevertheless, the relevant time scale for electron-lattice equilibration is in the order of 1 ps in  $\text{CrO}_2$  as well as in nickel. Therefore, the magnetic in-plane anisotropy constant undergoes a fast change in magnitude on the same time scale, which results in an anisotropic field pulse. The original configuration recovers, accompanied by precessional motion, on a time scale of 100 ps [Fig. 1(c)]. The measured transients for a pump fluence of  $F = 10 \text{ mJ/cm}^2$  are depicted in Fig. 2; it should be noted that these transients portray temporal changes of the value of the Kerr rotation as well as in the reflectivity. The transient reflectivity always adds with a positive value regardless of the applied field direction. For each field value, the two curves for both external field directions are taken directly after each other. By increasing the external field, the static configuration changes from transverse to longitudinal. Assuming the magnetization is rotated within the upper part of the full in-plane circle and the effective magnetic field pulse is always directed toward the hard axis, the magnetization is pulled above the film plane for a small positive field, while for a small negative field the magnetization is pulled below the film plane, resulting in a phase change of the transients for the polar and longitudinal component. For high fields, due to small imperfections always present in the alignment, the sense of rotation on switching the field direction is defined depending on the nearest easy axis. After laser excitation, the effective magnetic field pulse is directed into the same sense of rotation. The magnetization is therefore pulled in the same direction out of plane for both positive and negative applied fields. This change of symmetry is found in the transient oscillation for the high-field range. These transients can be perfectly

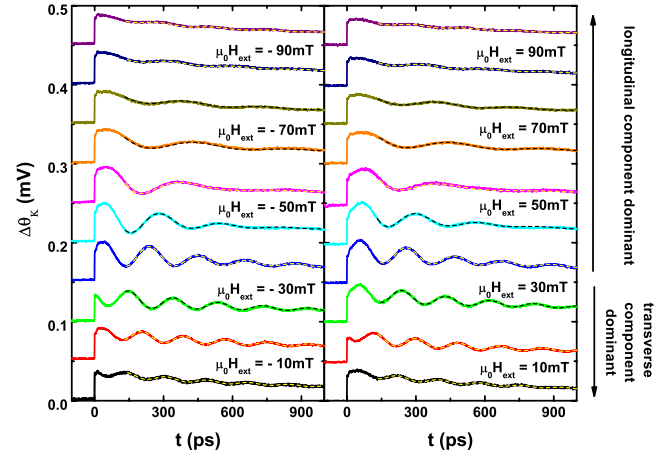


FIG. 2. (Color online) Changes in the reflectivity and in the real part of the Kerr rotation; fits are given by the dashed lines. The regions where prior to excitation the static component is dominantly transverse (for low fields) or longitudinal (for high fields) are marked. The pump fluency is  $F = 10 \text{ mJ/cm}^2$ . The curves are shifted for clarity.

fitted with functions of the type  $A \exp(-t/T) + B \exp(-t/\tau) \sin(\omega t + \phi)$ , where the first term portrays the change in reflectivity and a small change in the length of the Kerr vector. The second one represents a solution of the linearized LLG equation [Eq. (1)]. In the framework of this linearization, which corresponds to a parabolic approximation of the free energy, the precessional frequency  $\omega = 2\pi f$  and the damping time  $\tau$  can be expressed by the sample parameters  $M_s$ ,  $K_{1,\text{eff}}$ ,  $\alpha_{\text{eff}}$ , and the Landé factor  $g$  as well as the external field  $H_{\text{ext}}$  (neglecting quadratic terms in  $\alpha_{\text{eff}}$ ):

$$\omega = \frac{\gamma_0}{\mu_0 M_s \sin \theta_0} \sqrt{F_{\theta\theta} F_{\phi\phi} - F_{\theta\phi}^2},$$

$$\tau = \frac{2\mu_0 M_s}{\alpha \gamma_0 \left( F_{\theta\theta} + \frac{1}{\sin^2 \theta_0} F_{\phi\phi} \right)}, \quad (3)$$

where  $F_{ij}$  denotes the coefficients of the parabolic approximation of the free energy around the minimum. In Fig. 3(a), the extracted precessional frequency is plotted as a function of the external field for pump fluences of  $F = 10$  and  $20 \text{ mJ/cm}^2$ . The overall reduced frequency for the higher fluency can be explained by the higher average heating, which is also manifest in a reduction of the Kerr signal at negative delay times. The measured frequencies are fitted by Eq. (3), where, to avoid ambiguities, the Landé factor and the magnetization are held fixed at  $g = 2$  (Ref. 25) and  $\mu_0 M_s = 603 \text{ mT}$ .<sup>32</sup> The determined values are  $K_{1,\text{eff}} = 15\,990(75) \times 10^3 \text{ J/m}^3$  [ $15\,350(150) \times 10^3 \text{ J/m}^3$ ] for  $F = 10 \text{ mJ/cm}^2$  ( $20 \text{ mJ/cm}^2$ ).<sup>33</sup> The drop of the frequency at about  $\mu_0 H_{\text{ext}} = 65 \text{ mT}$  can be understood easily: the precessional motion of the magnetization means (in first order) a harmonic oscillation around the minimum of the free energy landscape of the sample system [Fig. 3(b)]. At external fields below the anisotropy field  $\mu_0 H_{\text{aniso}} = 2K_{1,\text{eff}}/M_s$ , there are

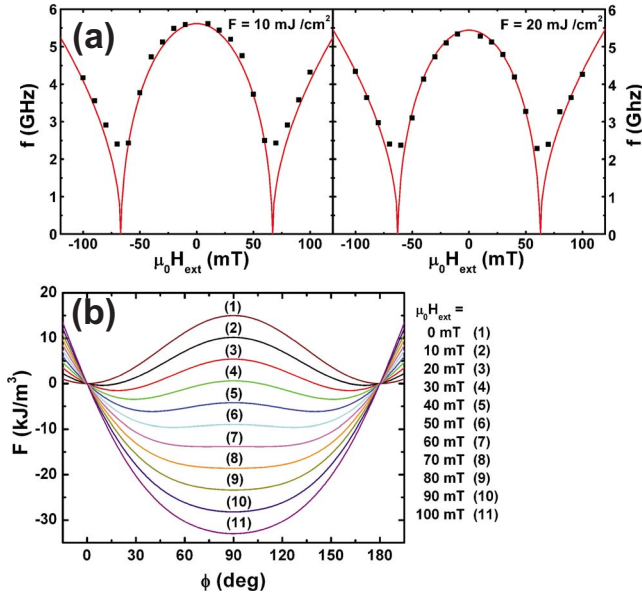


FIG. 3. (Color online) (a) Measured precessional frequency as a function of the external field for pump fluences  $F=10$  and  $20 \text{ mJ/cm}^2$  and fits according to Eq. (3). (b) Free energy density as a function of the azimuthal angle for different external field values.

two degenerate minima, which move toward each other with increasing field. At external fields above the anisotropy field, the magnetization is, of course, aligned with the external field. This transition from one to two minima at  $H_{\text{ext}}=H_{\text{aniso}}$  must be accompanied by vanishing second-order derivatives at the minimum position and, thus, vanishing restoring forces for the precessional motion as well.

Also, the damping time  $\tau$  is extracted from the measured Kerr transients and plotted for the two pump fluences in Fig. 4. The damping parameter  $\alpha_{\text{eff}}$  is calculated from  $\tau$  and the above determined sample parameters according to Eq. (3) and is included in Fig. 4. The damping shows a strong increase with decreasing precessional frequency as was also observed in similar experiments,<sup>17,19,20</sup> so that genuine Gil-

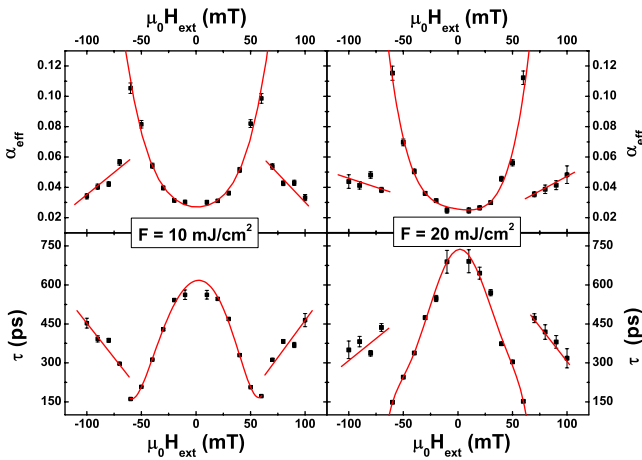


FIG. 4. (Color online) Field dependence of the damping time and the effective Gilbert damping parameter for fluences  $F=10$  and  $20 \text{ mJ/cm}^2$ ; the lines are guides for the eyes.

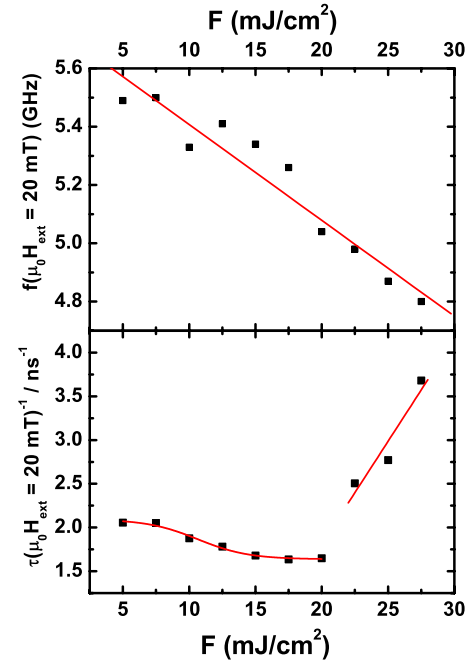


FIG. 5. (Color online) Pump fluence dependence of the precessional frequency and the inverse damping time for  $\mu_0 H_{\text{ext}}=20 \text{ mT}$ ; the lines are guides for the eyes. The energy dissipation rate increases sharply above the threshold value of  $F=20 \text{ mJ/cm}^2$ .

bert damping cannot be the dominant process of energy dissipation operative in this sample. For  $\text{CrO}_2$  films, an increase of the damping for lower precession frequencies was likewise found in ferromagnetic resonance (FMR).<sup>25</sup> In addition, these FMR measurements revealed that the linewidth in the in-plane hard-axis direction is broadened compared to that in the easy-axis direction. Thus, the damping in this sample might be not only frequency dependent, but also direction dependent. Woltersdorf and Heinrich demonstrated that the magnetic damping of an iron thin film on GaAs due to two-magnon scattering is enhanced in the directions of higher misfit dislocation density.<sup>26</sup> In the  $\text{CrO}_2$  film, there might be a higher misfit dislocation density in the  $b$ -axis direction compared to the  $c$ -axis direction due to the lattice mismatch anisotropy with the  $\text{TiO}_2$  substrate.<sup>27</sup> Qualitatively, transmission electron microscopy reveals that the strain is relieved by formation of misfit dislocations that stretch to the surface of the film; a quantitative study of the supposed dislocation density anisotropy is not possible. Another striking feature of the curves in Fig. 4 is the qualitatively different field dependence of the damping in the field range above  $\mu_0 H_{\text{ext}}=60 \text{ mT}$  for the two pump fluences. Thus, the pump fluence is an important parameter in this all-optical measurement. In Fig. 5, the frequency and the inverse damping time  $\tau^{-1}$  are plotted as functions of the laser fluence for an applied external field of  $\mu_0 H_{\text{ext}}=20 \text{ mT}$ . The frequency decreases with increased fluence as the equilibrium temperature rises due to the increased average heating. In the room temperature regime up to  $350 \text{ K}$ , according to FMR results,<sup>25</sup> the strength of damping is expected to decrease slightly with increasing temperature. This behavior is observed in our experiment up

to pump fluences of  $F=20 \text{ mJ/cm}^2$ ; above this threshold, there is a sharp increase of the damping. By measuring the length of the Kerr vector in comparison with superconducting quantum interference device measurements, it is verified that the spin temperature remains below 330 K for all fluences and time delays. Therefore, the increase of damping cannot be attributed to the increase of the spin temperature consistently with the FMR results. We suggest that this additional damping stems from spin-wave instabilities that have long been known to occur above a threshold in high-field-pumping FMR.<sup>28</sup> There, because of the high angle of excitation, the uniform precession becomes unstable due to

spin-wave disturbances and energy of the uniform mode is transferred to nonuniform ones. In time-resolved FMR, where the precession is excited by a single field pulse, these instabilities have been observed only in certain configurations.<sup>29,30</sup> The question of which peculiarities of the optically induced field pulse or the  $\text{CrO}_2$  sample become manifest in the spin-wave instabilities, suggested in this paper, remains open for future research.

The authors gratefully acknowledge support by the Deutsche Forschungsgemeinschaft within the priority program SPP 1133.

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<sup>1</sup>K. Schwarz, J. Phys. F: Met. Phys. **16**, L211 (1986).

<sup>2</sup>Y. S. Dedkov, M. Fonine, C. König, U. Rüdiger, G. Güntherodt, S. Senz, and D. Hesse, Appl. Phys. Lett. **80**, 4181 (2002).

<sup>3</sup>A. Anguelouch, A. Gupta, G. Xiao, D. W. Abraham, Y. Ji, S. Ingvarsson, and C. L. Chien, Phys. Rev. B **64**, 180408(R) (2001).

<sup>4</sup>Y. Ji, G. J. Strijkers, F. Y. Yang, C. L. Chien, J. M. Byers, A. Anguelouch, G. Xiao, and A. Gupta, Phys. Rev. Lett. **86**, 5585 (2001).

<sup>5</sup>W. E. Pickett and J. S. Moodera, Phys. Today **54**(5), 39 (2001).

<sup>6</sup>S. A. Wolf, D. D. Awschalom, R. A. Buhrman, J. M. Daughton, S. von Molnar, M. L. Roukes, A. Y. Chtchelkanova, and D. M. Treger, Science **294**, 1488 (2001).

<sup>7</sup>G. Ju, A. Vertikov, A. V. Nurmikko, C. Canady, G. Xiao, R. F. C. Farrow, and A. Cebollada, Phys. Rev. B **57**, R700 (1998).

<sup>8</sup>M. van Kampen, C. Jozsa, J. T. Kohlhepp, P. LeClair, L. Lagae, W. J. M. de Jonge, and B. Koopmans, Phys. Rev. Lett. **88**, 227201 (2002).

<sup>9</sup>E. Beaupaire, J.-C. Merle, A. Daunois, and J.-Y. Bigot, Phys. Rev. Lett. **76**, 4250 (1996).

<sup>10</sup>B. Koopmans, J. J. M. Ruigrok, F. Dalla Longa, and W. J. M. de Jonge, Phys. Rev. Lett. **95**, 267207 (2005).

<sup>11</sup>M. Djordjevic and M. Münzenberg, Phys. Rev. B **75**, 012404 (2007).

<sup>12</sup>C. Jozsa, J. H. H. Rietjens, M. van Kampen, E. Smalbrugge, M. K. Smit, W. J. M. de Jonge, and B. Koopmans, J. Appl. Phys. **95**, 7447 (2004).

<sup>13</sup>L. D. Landau and E. M. Lifschitz, in *L. D. Landau, Collected Papers*, edited by D. ter Haar (Pergamon, Oxford, 1965).

<sup>14</sup>T. L. Gilbert, IEEE Trans. Magn. **40**, 3443 (2004).

<sup>15</sup>M. Djordjevic, G. Eilers, A. Parge, M. Münzenberg, and J. S. Moodera, J. Appl. Phys. **99**, 08F308 (2006).

<sup>16</sup>Q. Zhang, A. V. Nurmikko, A. Anguelouch, G. Xiao, and A. Gupta, Phys. Rev. Lett. **89**, 177402 (2002).

<sup>17</sup>H. B. Zhao, D. Talbayev, Q. G. Yang, G. Lüpke, A. T. Hanbicki, C. H. Li, O. M. J. van 't Erve, G. Kioseoglou, and B. T. Jonker,

Appl. Phys. Lett. **86**, 152512 (2005).

<sup>18</sup>D. Talbayev, H. Zhao, G. Lüpke, A. Venimadhav, and Q. Li, Phys. Rev. B **73**, 014417 (2006).

<sup>19</sup>A. A. Rzhetsky, B. B. Krichevstov, D. E. Bürgler, and C. M. Schneider, Phys. Rev. B **75**, 224434 (2007).

<sup>20</sup>Y. Liu, L. R. Shelford, V. V. Kruglyak, R. J. Hicken, Y. Sakuraba, M. Oogane, Y. Ando, and T. Miyazaki, J. Appl. Phys. **101**, 09C106 (2007).

<sup>21</sup>G. Miao, G. Xiao, and A. Gupta, Phys. Rev. B **71**, 094418 (2005).

<sup>22</sup>F. Y. Yang, C. L. Chien, E. F. Ferrari, X. W. Li, G. Xiao, and A. Gupta, Appl. Phys. Lett. **77**, 286 (2000).

<sup>23</sup>X. W. Li, A. Gupta, and G. Xiao, Appl. Phys. Lett. **75**, 713 (1999).

<sup>24</sup>Q. Zhang, A. V. Nurmikko, G. X. Miao, G. Xiao, and A. Gupta, Phys. Rev. B **74**, 064414 (2006).

<sup>25</sup>P. Lubitz, M. Rubinstein, M. S. Osofsky, B. E. Nadgorny, R. J. Soulen, K. M. Bussmann, and A. Gupta, J. Appl. Phys. **89**, 6695 (2001).

<sup>26</sup>G. Woltersdorf and B. Heinrich, Phys. Rev. B **69**, 184417 (2004).

<sup>27</sup>G.-X. Miao, G. Xiao, and A. Gupta, Phys. Status Solidi A **203**, 1513 (2006).

<sup>28</sup>H. Suhl, J. Phys. Chem. Solids **1**, 209 (1957).

<sup>29</sup>T. J. Silva, P. Kabos, and M. R. Pufall, Appl. Phys. Lett. **81**, 2205 (2002).

<sup>30</sup>T. Gerrits, M. L. Schneider, A. B. Kos, and T. J. Silva, Phys. Rev. B **73**, 094454 (2006).

<sup>31</sup>H. B. Callen and E. Callen, J. Phys. Chem. Solids **27**, 1271 (1966).

<sup>32</sup>Therewith, the total temperature dependence of the sample due to the average heating is included in the anisotropy constant, which should exhibit the strongest temperature dependence (Ref. 31).

<sup>33</sup>The good agreement between measured and calculated frequencies also justifies the nonconsideration of the fact that the length of the magnetization vector is not conserved in the experiment due to the slow laser-induced demagnetization. For all pump fluences used in this work, the length of the magnetization vector (compared to the negative pump probe delay times) is reduced by less than 4%.